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The Crystal Structure of Wittichenite, Cu₃BiS₃*

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(Received 17 April 1973; accepted 5 June 1973)

Wittichenite, Cu₃BiS₃, from the type locality in Wittichen, West Germany is orthorhombic, $a = 7.723$ (10), $b = 10.395$ (10), $c = 6.716$ (5) Å; $D_{\text{meas}} = 6.01$, $D_{\text{calc}} = 6.11 \text{ g cm}^{-3}$; $Z = 4$; space group $P2_12_12_1$. The structure was solved by the heavy-atom method from 695 observed symmetry-independent reflexions $F^2_0 \geq 2\sigma F^2_0$ collected on a Picker four-circle FACS-1 diffractometer using Zr-filtered Mo $K\alpha$ ($\lambda = 0.71069$ Å) radiation. Spherical absorption corrections were applied to the data ($\mu R = 10.5$) and the structure was refined by full-matrix least-squares using anisotropic temperature factors to an R value of 4.9% ($R_w = 6.7\%$). The structure consists of infinite BiCu₃S₃ chains parallel to [001] which are linked by Cu–S bonds to form continuous sheets normal to [010]. Adjacent sheets are related by the 2_1 axes parallel to [100] and are linked by Cu–S and Bi–S bonds. Cu is in nearly trigonal planar coordination with S (Cu–S distances 2.255 to 2.348 Å; S–Cu–S angles 110.8° to 131.8°). The packing of CuS₃ polyhedra in the structure yields short Cu–Cu contacts ranging from 2.61 to 2.94 Å. Bi is trigonally coordinated by S (Bi–S distances 2.569 to 2.608 Å; S–Bi–S angles 94.2° to 98.7°). S is tetrahedrally coordinated by three copper atoms and one bismuth atom.

Introduction

Wittichenite is a bismuth sulphosalt of copper, apparently first described in 1805 by Selb (Hintze, 1904)

from Wittichen in Baden, Germany, under the name Kupferwismutherz. The original chemical analysis, which established the composition as Cu₃BiS₃, was performed by Klapproth in about 1807. Later the mineral was renamed wittichenit and finally wittichenite (wittichenite) in 1853.

The early mineralogists recognized a second copper bismuth sulphosalt in the ores from Wittichen, namely

* A report of this work was presented at the American Crystallographic Association meeting, Florida, January 1973, Paper H2.

kupferwismuthglanz, later re-named emplektit (emplectite), with composition CuBiS₂. A third name, klaprothite (or klaprotholite), thought to represent a compound of composition Cu₆Bi₄S₈, arose from investigations of the ores from the Daniel Mine in Wittichen. Detailed study (Nuffield, 1947) has shown that specimens labelled klaprothite from Wittichen are mixtures of wittichenite and emplectite and examination of specimens labelled klaprothite from elsewhere have not confirmed the existence of klaprothite.

A morphological and X-ray study of a well-developed crystal of wittichenite by Nuffield (1947) established the space group as $P2_12_12_1$ and provided the cell dimensions $a=7.68$, $b=10.33$, $c=6.70$ Å. Material synthesized by fusion of the elements in the course of the study gave a measured density of 6.19, compared with 6.01 measured by Frondel (Dana, 1944) on natural wittichenite labelled klaprothite, and 6.19 calculated for the cell with 4[Cu₃BiS₃].

Wittichenite has a metallic lustre. It breaks with a good conchoidal fracture. No cleavage has been observed. Face development is rare. The morphological habit is prismatic parallel to [001].

Experimental

The present study was undertaken on a specimen from the Daniel Mine in the type locality of Wittichen (Royal Ontario Museum specimen M 23304) on which a few rough, short-prismatic and heavily striated crystals of wittichenite were identified. An electron microprobe analysis of material selected from this specimen gave values which are in excellent agreement with the ideal composition, Cu₃BiS₃ (Table 1).

Table 1. Electron microprobe analysis of wittichenite from the Daniel Mine, Wittichen

		Atomic proportions	
	wt. %	Actual	Ideal
Cu	37.79	2.98	3
Bi	42.56	1.02	1
S	19.13	2.99	3
Total	99.48		

Analyst: Elvira Gasparrini.

Because of the irregular shape of the crystals and the high value of the linear absorption coefficient (440.2 cm⁻¹ for Mo K α), it was decided to grind spheres and apply spherical absorption corrections. The crystal that was finally used for data collection had a mean diameter of 0.46 mm and was spherical to within 0.04 mm of this value. The main deviation from sphericity was between the poles which were slightly flattened due to remnant {001} faces.

Lattice parameters were calculated from a number of high- θ reflexions measured on zero-layer Weissenberg and precession films, calibrated with copper powder lines and extrapolated to $\theta=90^\circ$ against the absorption function of Nelson & Riley (1945):

$$a=7.723(10), \quad b=10.395(10), \quad c=6.716(5) \text{ \AA}.$$

The calculated density for the new cell dimensions and the ideal cell contents 4[Cu₃BiS₃], is 6.11 g cm⁻³.

Intensity data were collected on a Picker four-circle FACS-1 diffractometer in the $\theta-2\theta$ scan mode up to 55° in 2 θ , using Zr-filtered Mo K α ($\lambda=0.71069$ Å) radiation. A scan rate of 0.5° min⁻¹ with a 40 s background count on each side of the peak was used to improve the counting statistics of the weaker reflexions. Copper foil attenuators were automatically inserted into the diffracted beam whenever the counting rate exceeded 10000 c.p.s. The intensities of two standard reflexions were monitored every 40 reflexions. The variation throughout the data collection was less than 1.5%.

In total, 722 symmetry-independent reflexions were collected. The data were scaled by reference to the standard reflexions and corrected for Lorentz and polarization factors. The spherical absorption factors calculated by Weber (1969) for a μR value of 10.5 were applied to the data. A total of 695 intensities had $|F|^2 \geq 2\sigma|F|^2$.

Determination and refinement of the structure

The position of Bi was obtained from a three-dimensional Patterson map. The remainder of the structure was solved by conventional heavy-atom methods.

Structure factor calculations with the Bi position gave $R=0.39$, where R is defined as: $\sum ||F_o|| - |F_c|| / \sum |F_o||$. A subsequent F_o Fourier map revealed one Cu and the three S positions. The other two Cu positions were

Table 2. Fractional coordinates and temperature factors

Fractional coordinates and anisotropic thermal parameters $\times 10^5$; values in parentheses are estimated standard deviations as obtained from the last cycle of least-squares refinement;

$$T = \exp [-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl)].$$

	x	y	z	β_{11}	β_{22}	β_{33}	β_{12}	β_{13}	β_{23}	B (Å ²)
Bi	19716 (10)	-24401 (7)	12860 (12)	230 (15)	314 (8)	856 (20)	-20 (8)	33 (12)	-2 (10)	1.24
Cu(1)	35179 (52)	40167 (37)	13762 (56)	852 (64)	689 (34)	1069 (74)	346 (36)	22 (60)	41 (47)	2.10
Cu(2)	19037 (84)	10470 (43)	7838 (60)	2433 (113)	720 (40)	914 (72)	590 (61)	312 (88)	234 (47)	2.75
Cu(3)	5727 (44)	46445 (35)	-2628 (59)	403 (53)	487 (31)	1552 (82)	66 (32)	257 (60)	-36 (43)	1.87
S(1)	-44584 (78)	24619 (57)	10918 (83)	433 (90)	364 (47)	705 (109)	8 (56)	-164 (82)	25 (73)	1.32
S(2)	-31767 (82)	43669 (56)	-40779 (92)	325 (85)	338 (44)	752 (126)	52 (55)	8 (105)	-36 (60)	1.47
S(3)	-32136 (83)	6949 (53)	-41380 (86)	370 (87)	276 (42)	626 (120)	23 (52)	2 (100)	-78 (56)	1.49

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located from a second Fourier map calculated with refined phases. Four cycles of *XFLS* (Ellison, 1962) isotropic least-squares refinement with all atoms in the structure-factor calculation reduced the *R* value to

0.08. At this stage corrections for anomalous dispersion were introduced using the values computed by Cromer (1965). Two cycles of anisotropic least-squares refinement brought the *R* value to 0.05. The scattering

Table 3. Comparison of observed and calculated structure factors listed as 10*F*_o and 10*F*_c

Unobserved values indicated by an asterisk. Phase angles are in millicycles.

L	FO	FC	ALPHA	L	FO	FC	ALPHA	L	FO	FC	ALPHA	L	FO	FC	ALPHA	L	FO	FC	ALPHA	L	FO	FC	ALPHA	L	FO	FC	ALPHA								
Hm	O ₁	K ₀	0	Hm	I ₁	K ₀	0	2	799	798	557	0	482	422	568	6	351	325	816	1	243	241	470	Hm	I ₀	K ₁	1	4	214	252	161				
Z	1567	1669	5	1	919	195	329	3	295	292	155	1	912	705	253	2	777	722	816	3	153	163	251	0	1032	1005	51	5	607	590	240				
4	3008	3780	517	2	2700	213	302	5	226	220	104	2	264	204	259	3	795	782	259	Hm	I ₃	K ₁₁	11	1	718	831	706	Hm	I ₇	K ₅	5				
6	468	429	491	3	226	213	302	3	180	182	13	3	793	782	259	6	655	635	555	0	415	389	280	2	378	356	607	0	131	87	315				
8	1043	1020	39	4	204	19	9	0	346	355	300	Hm	I ₂	K ₁₂	12	1	390	398	246	0	68	59	109	3	37	365	325	1	609	701	317				
Hm	O ₂	K ₁	1	5	192	193	814	1	301	605	224	Hm	I ₂	K ₁₂	12	2	1058	1073	58	1	638	593	763	5	592	577	377	2	1181	1270	549				
1	779	812	796	8	377	358	516	Hm	I ₂	K ₁₂	12	3	841	802	541	3	292	308	*367	0	301	316	63	3	652	598	270	9	256	217	63				
2	455	250	265	Hm	I ₁	K ₀	0	0	2577	2711	520	3	543	513	875	Hm	I ₃	K ₁₂	12	1	2270	2016	274	Hm	I ₀	K ₂	2	4	214	252	161				
3	1361	1426	282	Hm	I ₁	K ₀	1	0	2577	2711	520	1	132	1364	0	2	1050	965	16	0	447	463	556	4	7	K ₀	6								
5	938	714	521	0	1426	181	750	5	192	193	746	Hm	I ₂	K ₁₃	13	0	670	648	783	2	2253	1945	978	1	427	424	292	1	531	501	797				
6	603	615	262	1	1210	1340	319	3	202	113	31	1	679	656	802	4	103	105	30	1	234	143	326	0	382	433	278								
7	817	818	794	2	1602	1639	29	4	2266	2241	24	0	594	570	38	2	455	421	19	5	884	838	795	3	1526	1413	267	1	621	606	217				
8	708	B6	767	3	1008	1248	421	5	304	296	014	1	417	384	361	6	199	195	522	4	316	225	853	2	038	726	119	7	766	608	282				
Hm	O ₃	K ₂	2	5	1555	1317	721	2	343	382	292	Hm	I ₃	K ₀	0	1	766	608	282	5	1126	1105	311	3	609	588	270	0	410	442	010				
6	860	884	612	8	495	520	555	Hm	I ₂	K ₀	0	1	133	1364	0	2	292	243	74	1	340	281	898	Hm	I ₀	K ₂	2	5	612	620	629				
3	1361	1426	282	Hm	I ₁	K ₀	1	0	2577	2711	520	3	543	513	875	Hm	I ₃	K ₁₂	12	1	2270	2016	274	Hm	I ₀	K ₂	2	4	7	K ₀	6				
5	138	714	521	0	1426	181	750	5	192	193	746	Hm	I ₂	K ₁₃	13	0	670	648	783	2	2253	1945	978	1	427	424	292	1	531	501	797				
6	603	615	262	1	1210	1340	319	3	202	113	31	1	679	656	802	4	103	105	30	1	234	143	326	0	382	433	278								
7	817	818	794	2	1602	1639	29	4	2266	2241	24	0	594	570	38	2	455	421	19	5	884	838	795	3	1526	1413	267	1	621	606	217				
8	708	B6	767	3	1008	1248	421	5	304	296	014	1	417	384	361	6	199	195	522	4	316	225	853	2	038	726	119	7	766	608	282				
Hm	O ₄	K ₂	2	5	1555	1317	721	2	343	382	292	Hm	I ₃	K ₀	0	1	766	608	282	5	1126	1105	311	3	609	588	270	0	410	442	010				
1	779	812	796	8	495	520	555	Hm	I ₂	K ₀	0	1	133	1364	0	2	292	243	74	1	340	281	898	Hm	I ₀	K ₂	2	5	612	620	629				
2	1106	1133	10	8	143	169	777	Hm	I ₂	K ₀	1	0	2577	2711	520	3	543	513	875	0	268	254	763	Hm	I ₀	K ₂	2	4	7	K ₀	6				
3	385	395	505	Hm	I ₁	K ₀	2	0	1054	1137	300	3	391	300	408	4	210	197	74	0	670	648	783	2	2253	1945	978	1	427	424	292	1	531	501	797
4	203	2172	282	Hm	I ₁	K ₀	2	0	1054	1137	300	3	391	300	408	5	207	196	74	0	670	648	783	2	2253	1945	978	1	427	424	292	1	531	501	797
5	625	522	572	0	1426	181	750	5	192	193	746	Hm	I ₂	K ₁₃	13	0	670	648	783	2	2253	1945	978	1	427	424	292	1	531	501	797				
6	253	229	524	1	130	435	722	5	150	142	672	5	172	70	278	7	014	644	792	5	137	126	204	3	194	181	204	5	1234	1208	204				
7	112	84	534	2	2437	2431	7	1	160	123	542	5	110	91	49	Hm	I ₄	K ₀	1	7	72	140	653	Hm	I ₇	K ₈	8	7	72	140	653	Hm	I ₇	K ₈	8
8	1035	1026	537	3	657	587	591	0	859	858	786	Hm	I ₃	K ₀	2	0	108	80	160	Hm	I ₅	K ₁	1	0	195	231	791	Hm	I ₈	K ₁	1	0	195	231	791
Hm	O ₅	K ₃	3	5	351	285	184	7	710	709	292	Hm	I ₃	K ₀	2	0	2393	2413	527	Hm	I ₅	K ₂	2	Hm	I ₆	K ₄	4	0	195	231	791				
1	6177	1184	550	8	613	739	89	0	1418	1337	760	1	689	610	887	Hm	I ₃	K ₀	2	0	2393	2413	527	Hm	I ₆	K ₄	4	1	529	618	808				
2	1232	1200	505	7	608	571	964	Hm	I ₁	K ₀	3	0	1596	1602	595	2	160	161	595	0	246	250	250	6	115	100	284	2	1013	1033	497				
3	1534	1545	780	8	149	155	994	Hm	I ₂	K ₀	2	0	2393	2413	527	3	1620	1639	595	1	1420	1244	207	3	556	563	891								
4	81	8	342	Hm	I ₁	K ₀	3	0	1753	1808	300	4	401	390	990	5	148	147	300	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
5	651	558	804	0	109	208	29	3	144	1416	521	5	149	140	493	6	109	92	671	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
6	1137	1138	282	7	608	571	964	Hm	I ₂	K ₀	2	0	2393	2413	527	1	1574	1604	204	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
7	1193	1203	781	8	1053	1028	23	Hm	I ₂	K ₀	2	0	209	217	269	2	1357	1426	239	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
8	563	599	273	7	213	205	914	Hm	I ₂	K ₀	2	1	857	825	236	3	222	220	524	1	1307	1226	238	2	337	327	770	Hm	I ₆	K ₄	4	0	195	231	791
9	513	512	519	3	503	518	578	Hm	I ₁	K ₀	3	0	1393	1361	527	1	1443	1416	301	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
10	575	572	575	3	503	518	578	Hm	I ₂	K ₀	3	0	1393	1361	527	1	1443	1416	301	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
11	575	572	575	3	503	518	578	Hm	I ₃	K ₀	3	0	1393	1361	527	1	1443	1416	301	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
12	575	572	575	3	503	518	578	Hm	I ₄	K ₀	3	0	1393	1361	527	1	1443	1416	301	0	2393	2413	527	Hm	I ₆	K ₄	4	0	195	231	791				
13	575	572	575	3	503	518	578	Hm	I ₅	K ₀	3	0	1393	1361	527	1	1443	1416</td																	

factors used in these calculations were taken from Cromer & Mann (1968).

The initial weights of the structure factors were derived from the counting statistics as described by Stout & Jensen (1968). However, the standard deviation of an observation of unit weight calculated from the expression

$$S_1 = [\sum w \Delta F^2 / (m - n)]^{1/2}$$

in which w is the weight, m the number of observations and n the number of parameters varied in the least-squares refinement, had a value of 2.52 indicating slight errors in the assignment of the weights to the structure factors. A plot of \bar{F}_o/F_o vs. $1/\bar{F}_o$ for groups of 50 reflexions was fitted by a least-squares routine to a second order polynomial from which the expression $(0.039F_o + 1.30 + 24.37/F_o)$ was derived for σ_{F_o} and used in the final weighting scheme, $w = 1/\sigma_{F_o}^2$. Extra cycles of anisotropic refinement with new weights converged to $R = 0.049$ and $R_w = [\sum w(|F_o| - |F_c|)^2 / \sum w|F_o|^2]^{1/2} = 0.067$. The final value of S_1 equalled 1.06.

A difference Fourier map revealed an excess electron density of about $2.8 \text{ e } \text{\AA}^{-3}$ in the vicinity of the Bi atom, the actual centre of the atom being located at $+1.5 \text{ e } \text{\AA}^{-3}$. The remainder of the map did not reveal any anomalies.

The atomic positional parameters and their corresponding temperature factors are listed in Table 2. The comparison of observed and calculated structure factors is given in Table 3.

An attempt to establish the absolute configuration of the structure (Ibers & Hamilton, 1964), resulted in $R = 0.048$ for the enantiomorph. The ratio R_1/R_2 is 1.02 and according to Hamilton's (1965) significance test, the value, $\mathcal{R}_{1,632,0.01} = 1.005$, indicates that the enantiomorph can be accepted as a correct model at a 0.01 significance level. However, the result of the test should be treated with some reservation in the present case. Because Bi is a strong anomalous scatterer ($Mo\text{K}\alpha$, $\Delta f' = -4.54$, $i\Delta f'' = 10.65$), one would expect a more pronounced difference between the two R values. The explanation for the small difference must lie in the

material itself. Wittichenite, like most sulphosalt minerals, has a non-uniform mosaic spread over the volume of the crystal. This is likely to produce systematic errors in the intensities of some reflexions. The configuration shown here (Table 2) is the one which corresponds to the higher R value.

Discussion of the structure

Tables 4 and 5 give, respectively, the interatomic distances and angles in the wittichenite structure and their estimated standard deviations as calculated from the ORFFE (Busing, Martin & Levy, 1964) program. The

Table 5. Interatomic angles

Estimated standard deviation of all angles is $0.5^\circ(2\sigma)$

Metal-atom polyhedra	Bi	$S(1^{iv})—Bi—S(2^{iv})$	98.7
		$S(1^{iv})—Bi—S(3^{iv})$	94.2
		$S(2^{iv})—Bi—S(3^{iv})$	95.2
	Cu(1)	$S(1^i) — Cu(1) — S(3^{iii})$	124.2
		$S(1^i) — Cu(1) — S(3^{iv})$	123.7
		$S(3^{iii}) — Cu(1) — S(3^{iv})$	111.4
	Cu(2)	$S(1^{iii}) — Cu(2) — S(2^{iii})$	131.8
		$S(1^{iii}) — Cu(2) — S(2^{iv})$	116.2
		$S(2^{iii}) — Cu(2) — S(2^{iv})$	110.8
	Cu(3)	$S(1^{iii}) — Cu(3) — S(2^{ii})$	121.2
		$S(1^{iii}) — Cu(3) — S(3^{iv})$	114.7
		$S(2^{ii}) — Cu(3) — S(3^{iv})$	124.0
Sulphur-atom polyhedra	S(1)	$Bi^{iv} — S(1) — Cu(1^i)$	115.2
		$Bi^{iv} — S(1) — Cu(2^{iii})$	90.4
		$Bi^{iv} — S(1) — Cu(3^{iii})$	101.3
		$Cu(1^i) — S(1) — Cu(2^{iii})$	83.0
		$Cu(1^i) — S(1) — Cu(3^{iii})$	136.3
		$Cu(2^{iii}) — S(1) — Cu(3^{iii})$	121.5
	S(2)	$Bi^{iv} — S(2) — Cu(2^{iii})$	114.7
		$Bi^{iv} — S(2) — Cu(2^{iv})$	96.5
		$Bi^{iv} — S(2) — Cu(3^{ii})$	115.2
		$Cu(2^{iii}) — S(2) — Cu(2^{iv})$	128.3
		$Cu(2^{iii}) — S(2) — Cu(3^{ii})$	117.0
		$Cu(2^{iv}) — S(2) — Cu(3^{ii})$	80.2
	S(3)	$Bi^{iv} — S(3) — Cu(1^{iii})$	97.8
		$Bi^{iv} — S(3) — Cu(1^{iv})$	103.8
		$Bi^{iv} — S(3) — Cu(3^{iv})$	86.1
		$Cu(1^{iii}) — S(3) — Cu(1^{iv})$	132.8
		$Cu(1^{iii}) — S(3) — Cu(3^{iv})$	72.2
		$Cu(1^{iv}) — S(3) — Cu(3^{iv})$	68.1

Table 4. Interatomic distances

Symmetry positions: i x, y, z ; ii $\frac{1}{2} - x, \bar{y}, \frac{1}{2} + z$; iii $\frac{1}{2} + x, \frac{1}{2} - y, \bar{z}$; iv $\bar{x}, \frac{1}{2} + y, \frac{1}{2} - z$.
Estimated standard deviations in parentheses.

Metal-atom polyhedra	Bi		Cu(1)		Cu(2)		Cu(3)	
	$S(1^{iv})$	$2.608 (6) \text{ \AA}$	$S(1^i)$	$2.256 (7) \text{ \AA}$	$S(1^{iii})$	$2.257 (7) \text{ \AA}$	$S(1^{iii})$	$2.259 (7) \text{ \AA}$
	$S(2^{iv})$	$2.569 (6)$	$S(3^{iii})$	$2.306 (7)$	$S(2^{iii})$	$2.255 (8)$	$S(2^{ii})$	$2.261 (7)$
	$S(3^{iv})$	$2.600 (6)$	$S(3^{iv})$	$2.314 (7)$	$S(2^{iv})$	$2.309 (7)$	$S(3^{iv})$	$2.348 (8)$
Sulphur-atom polyhedra	S(1)		S(2)		S(3)			
	Bi^{iv}	$2.608 (6) \text{ \AA}$	Bi^{iv}	$2.569 (6) \text{ \AA}$	Bi^{iv}	$2.600 (6) \text{ \AA}$		
	$Cu(1^i)$	$2.256 (7)$	$Cu(2^{iii})$	$2.255 (8)$	$Cu(1^{iii})$	$2.306 (7)$		
	$Cu(2^{ii})$	$2.257 (7)$	$Cu(2^{iv})$	$2.309 (7)$	$Cu(1^{iv})$	$2.314 (7)$		
	$Cu(3^{ii})$	$2.259 (7)$	$Cu(3^{ii})$	$2.261 (7)$	$Cu(3^{iv})$	$2.348 (8)$		
Shortest metal-metal distances	$Bi — Bi^{iii}$		$4.232 (1) \text{ \AA}$					
	$Cu(1) — Cu(3^i)$		$2.610 (5)$					
	$Cu(2) — Cu(3^{ii})$		$2.944 (8)$					
	$Cu(3) — Cu(1^i)$		$2.743 (6)$					

Table 6. Average Cu-S distances and average S-Cu-S angles in Cu-S₃ polyhedra, and deviations of the polyhedra from planarity

Average Cu-S distances	Average S-Cu-S angles	Atoms in the plane*	Atoms out of the plane	<i>A</i>	<i>B</i>	<i>C</i>	<i>D</i>
2.292 Å	119.8°	S(1 ^{III})†, S(3 ^{III}), S(3 ^{IV})	Cu(1) 0.101 Å	-0.692	-0.552	-0.465	4.717
2.274	119.6	S(1 ^{III}), S(2 ^{III}), S(2 ^{IV})	Cu(2) -0.150	-0.849	-0.526	-0.058	1.699
2.269	120.0	S(1 ^{III}), S(2 ^{II}), S(3 ^{IV})	Cu(3) 0.029	-0.290	0.226	-0.929	-1.157

* The equation of the plane is $Ax + By + Cz + D = 0$.

† The symmetry code is the same as in Table 4.

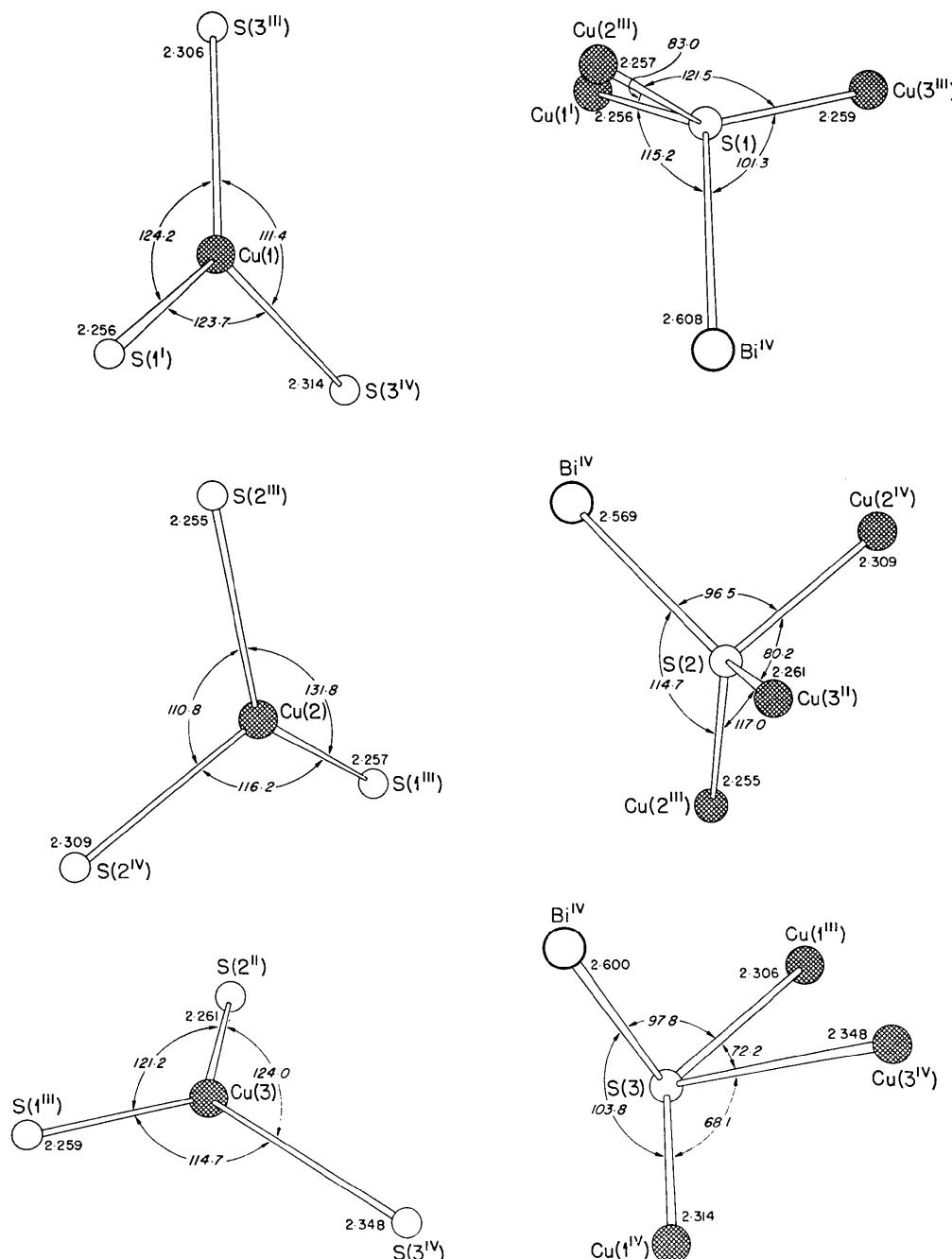


Fig. 1. Cu in trigonal planar coordination with S. Coordination of 3Cu and 1Bi about S.

standard deviations include the errors in cell parameters.

All Cu atoms are in nearly trigonal planar coordination with S. The Cu-S distances range from 2.255 to 2.348 Å and the S-Cu-S angles from 110.8 to 131.8° (Fig. 1). Table 6 gives the averages of these values and the offset of the Cu atom from the S plane for each of the three CuS_3 polyhedra. Although Cu is usually tetrahedrally coordinated by S in the ore minerals, trigonal coordination has been observed in a few structures including low-chalcite Cu_2S (Evans, 1971), the unnamed $\text{PbCu}_4\text{Bi}_5\text{S}_{11}$ (Kupčík & Makovický, 1968),

and the isostructural compounds binnite $\text{Cu}_{12}\text{As}_4\text{S}_{13}$ (Wuensch, Takeuchi & Nowacki, 1966) and tetratahedralite $\text{Cu}_{12}\text{Sb}_4\text{S}_{13}$ (Wuensch, 1964). The Cu-S distances in the CuS_3 polyhedra in these minerals vary between 2.20 and 2.89 Å with an average of about 2.28 Å. The wittichenite values are, therefore, normal.

The Bi atom and its three closest S neighbours occupy the corners of a trigonal pyramid with Bi at the vertex (Fig. 2). The Bi-S bond lengths are 2.569, 2.600 and 2.608 Å; the S-Bi-S angles are nearly orthogonal, ranging from 94.2 to 98.7°. More distant S neighbours occur at 3.434 Å and beyond this value (Table 7).

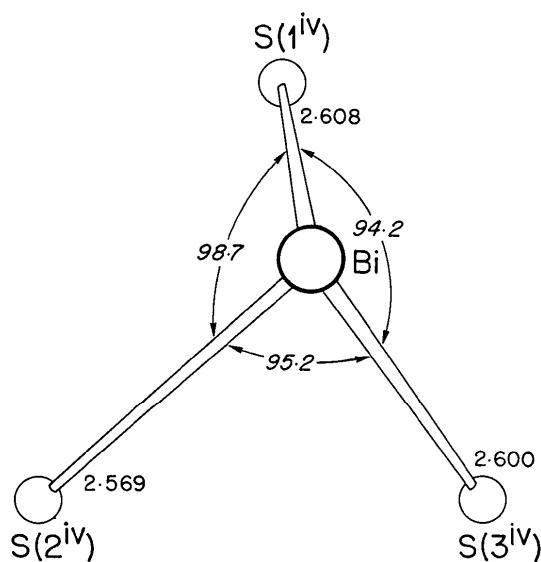


Fig. 2. Bi in trigonal pyramidal coordination with S.

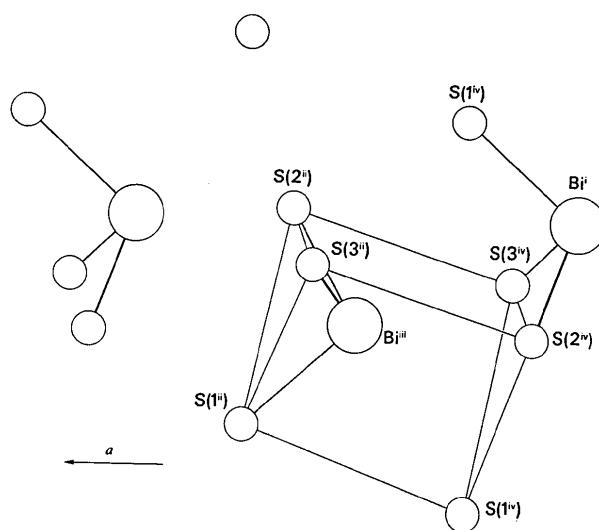


Fig. 3. Arrangement of Bi along [100]. $\text{Bi}-\text{Bi}=4.232$ Å. $\text{Bi}^{\text{III}}-\text{S}$ distances: $\text{S}(2^{\text{II}})$ 2.569, $\text{S}(3^{\text{II}})$ 2.600, $\text{S}(1^{\text{II}})$ 2.608, $\text{S}(3^{\text{IV}})$ 3.434, $\text{S}(2^{\text{IV}})$ 3.558, $\text{S}(1^{\text{IV}})$ 3.766, 3.992 Å.

Table 7. Closest S neighbours to Bi

Bi at $x=0.20$, $y=0.76$, $z=0.13$.

No.	S	z^*	Bi-S
1	2^{IV}	-0.09	2.569 Å†
2	3^{IV}	-0.09	2.600
3	1^{IV}	0.39	2.608
4	3^{II}	0.09	3.434
5	2^{II}	0.09	3.558
6	2^{III}	0.41	3.706
7	1^{II}	0.61	3.766
8	3^{III}	0.41	3.891
9	1^{I}	-0.39	3.992

* Fractional coordinates of S atoms.

† E.s.d. 0.006–0.007 Å.

Kohatsu & Wuensch (1971) have compared the coordination of Bi in the known bismuth ore mineral structures. Three short (2.7 ± 0.15 Å), nearly orthogonal Bi-S bonds may be identified in all the structures. Three or four more distant S neighbours are also present of which the fourth and fifth S atoms at distances of 3.0 ± 0.15 Å usually complete a square pyramid about the Bi atom. When the fourth and fifth S atoms are taken into account, a square pyramid of S atoms about Bi can also be observed in wittichenite. However, the coordination of S about Bi in wittichenite has certain unique features. Firstly, the three nearest S atoms are closer to the Bi atom than in the other structures and secondly, the first of the more distant neighbours is at 3.43 Å, rather than at about 3.0 Å. The Bi coordination in wittichenite is, therefore, more rigorously trigonal and resembles the coordination which is characteristic for S about As and to a lesser extent about Sb, in the sulphosalts in which these semi-metals prevail.

The shortest Bi-Bi contacts (4.232 Å) are in the (010) plane in which Bi is arranged in a zigzag fashion parallel to [100]. The BiS_3 polyhedra along such a row point in the same direction and may be pictured as loosely connected into a chain in which each Bi atom shares two of its three closest S atoms with one other Bi atom (Fig. 3). The coordinating S polyhedra have the form of trigonal prisms which are joined by sharing an edge in the basal plane. In such a prism, the Bi atom has three close and three considerably more distant S neighbours (atoms 4, 5 and 7 of Table 7) at 3.434, 3.558 and 3.766 Å.

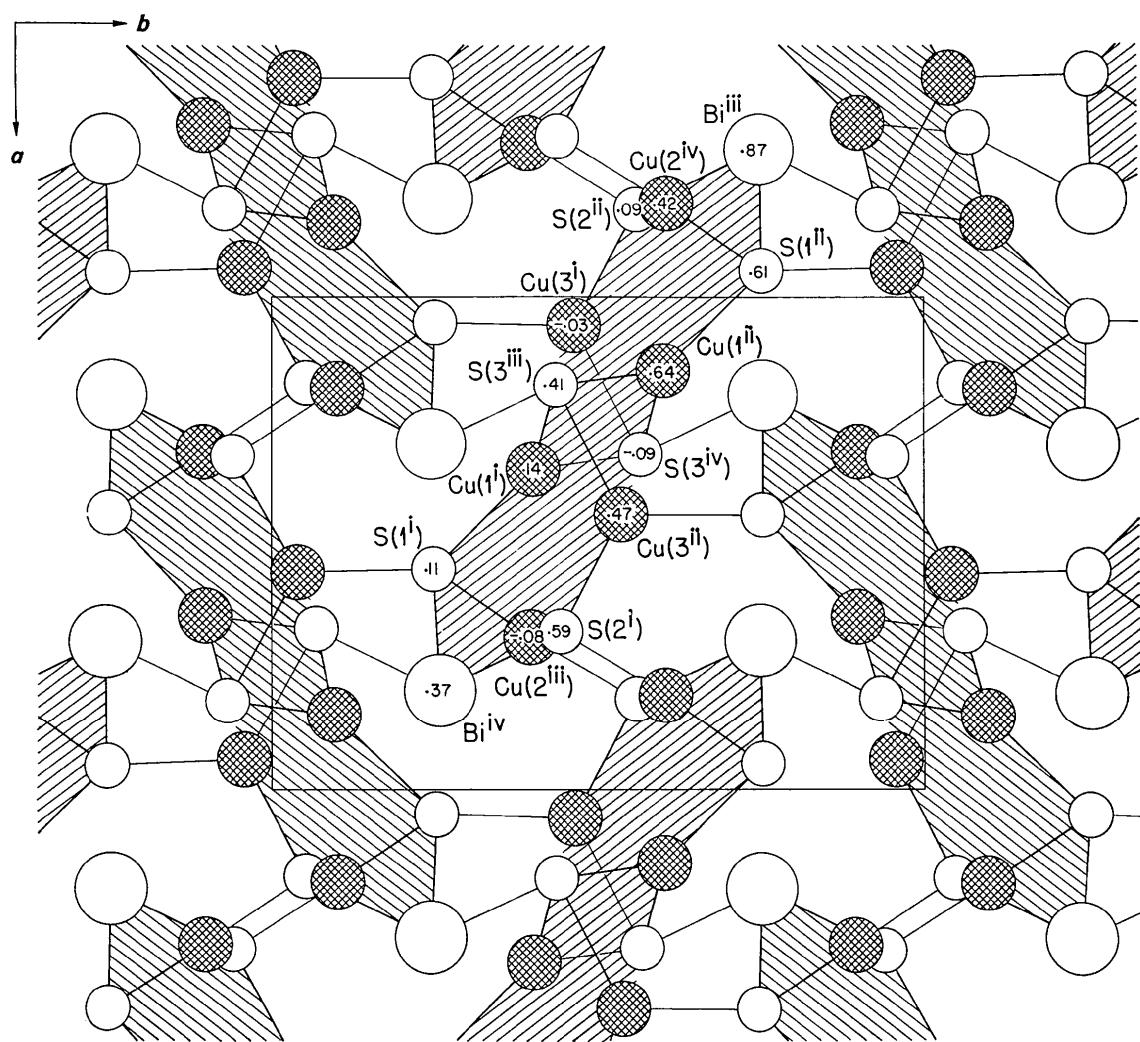


Fig. 4. Projection of the wittichenite structure on (001). Fractional coordinates are given for one chain.

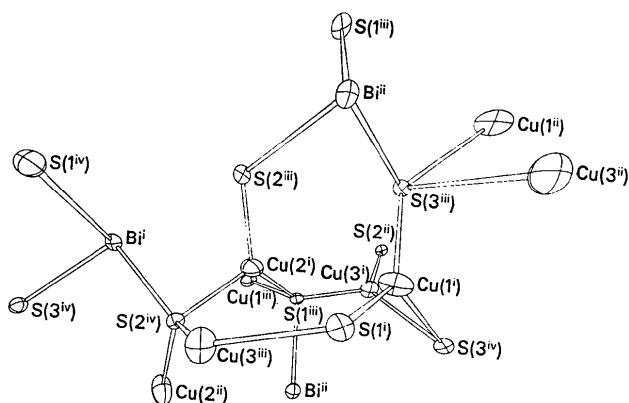


Fig. 5. ORTEP plot of part of wittichenite structure showing anisotropic thermal vibrations of the atoms drawn with 50% probability.

Each S atom coordinates one Bi and three Cu atoms (Fig. 1). The metal-S-metal angles given in Table 5 indicate a wide departure from the ideal tetrahedral bond angle of $109^\circ 28'$. A comparison of the mean S-Bi and S-Cu bond lengths in wittichenite with the sums of the corresponding covalent, tetrahedral radii (Evans, 1964) shows that S-Bi distances are slightly longer and S-Cu distances slightly shorter than the sums.

Mean S-metal distances in wittichenite	Sum of tetrahedral covalent radii (Evans, 1964)
S-Bi 2.59 Å	2.50 Å
S-Cu 2.29	2.39

Fig. 4 is a projection of the structure of wittichenite on (001). The most prominent and distinct structural feature is a series of infinite BiCu_3S_3 chains extending

parallel to [001]. Each chain is symmetrical about a central 2_1 axis. The chains are situated with the intermediate dimension, which is about 10.4 Å in length, parallel to (120) or (1 $\bar{2}$ 0). Parallel chains are linked by Cu-S bonds at intervals of 3.36 Å along [001] and overlap slightly to form continuous sheets normal to [010]. Adjacent sheets are related by 2_1 axes parallel to [100] and linked by Cu-S and Bi-S bonds. The two types of bond alternate along [100] at intervals of $\frac{1}{2}a$ and occur at intervals of 1c along [001]. The unit cell contains two sheets.

The three kinds of Cu₃S₃ polyhedron have distinctly different functions in the structure. The Cu(1) polyhedra form the cores of the chains and lie entirely within individual chains. Only two S atoms of each Cu(2) polyhedron lie in the same chain; the third S is in an overlapping chain and therefore the Cu(2) polyhedra link the chains into sheets. Similarly, only two S atoms of each Cu(3) polyhedron are in the same chain; the third S is in the chain of an adjacent sheet. Thus the Cu(3) polyhedra assist in binding the sheets into a three-dimensional structure.

The structure of wittichenite bears a resemblance to other sulphosalts structures that consist of frameworks of complex, essentially discrete chains (e.g. aikinite, PbCuBiS₃), but the similarity is superficial. The wittichenite chains are basically Cu₃S₃ in composition and characterized by trigonal planar coordination of S about Cu (Fig. 5). The more common coordination in sulphosalts is tetrahedral.

We wish to express our thanks to Miss Elvira Gasparini for the electron microprobe analysis, to Dr J. Mandarino of the Royal Ontario Museum for the loan of a specimen, to Dr Klaus Dichmann for providing us with the weighting program and to the National Research Council for a grant (to E.W.N.) to support the work.

Note added in proof:—A short account of an independent solution of the structure of wittichenite has recently

been given by Matzat (1972). The refinement was carried to an *R* value of 13.5% with isotropic temperature factors. The interatomic distances and angles are in substantial agreement with our values.

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